

# Synthesis and characterization of mordenites encapsulated titania nanoparticles: Photocatalytic degradation of *meta*-chlorophenol

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## Abstract

Titanium-mordenites catalysts prepared by impregnation ( $\text{TiM}_{\text{imp}}$ ), solid-solid interaction ( $\text{TiM}_{\text{ss}}$ ) and in situ incorporation during synthesis ( $\text{TiM}_{\text{in}}$ ); using *ortho*-phenylenediamine as a new templating agent under hydrothermal conditions ( $\text{TiCl}_4$ , 160 °C, 6 days), were thoroughly characterized by TG/DTA,  $\text{N}_2$  adsorption, XRD, FTIR, UV–vis and pyridine-FTIR. They were tested; in comparison with  $\text{TiO}_2$  Degussa, for degradation of *meta*-chlorophenol (MCP) in the presence of UV-light irradiation. It was possible to correlate the results obtained for MCP degradation over various catalysts with their structural and acidic properties. The photocatalytic degradation of MCP has been modeled to Langmuir-Hinshelwood rate law.  $\text{TiM}_{\text{ss}}$  showed the highest degradation rate ( $0.060 \text{ min}^{-1}$ ) followed by those of  $\text{TiM}_{\text{imp}}$  ( $0.050 \text{ min}^{-1}$ ),  $\text{TiM}_{\text{in}}$  ( $0.025 \text{ min}^{-1}$ ) and  $\text{TiO}_2$ -Degussa ( $0.018 \text{ min}^{-1}$ ), respectively. The marked higher rate observed for the former catalyst was due to the larger pore radius (29 Å), acidity and the extra-framework  $\text{TiO}_2$  anatase species (of absorption at 340 nm). Very strong emphasize to the state of titania within the three samples ( $\text{TiM}$ ) was demonstrated and most importantly Si–O–Ti linkages ( $950\text{--}940 \text{ cm}^{-1}$ ) were observed and correlated with photoactivities of all samples. Degradation products of MCP, including both aromatic and aliphatic intermediates, were monitored by HPLC. Chloride ions, on the other hand, were followed up during the degradation processes. Formic acid (formate) was identified as a final intermediate in the solution before it further mineralized to  $\text{CO}_2$ . Reasonable degradation pathway for MCP molecule is proposed on the basis of structural identification of several by-products. © 2007 Elsevier B.V. All rights reserved.

**Keywords:** Mordenite; Synthesis; Ti-incorporation; Characterization; Photocatalytic degradation

## 1. Introduction

Mordenite is an industrially important zeolite used as solid catalyst for conversion of hydrocarbons and is recently being considered for applications as host of semiconductor materials, chemical sensors and non-linear optical materials [1–4]. Therefore, there are a lot of articles and patents concerning the synthesis of mordenite zeolite crystals in the absence or presence of organics such as piperazine and polyglycol [5–8]. Isomorphous substitution of a metal ion into the framework, on the other hand, of mordenite is very effective in exhibiting spectacular catalytic properties based on the acidity of the bifunctional metal ion–zeolite system [9–10].

A comprehensive study of the synthesis of mordenite using a new template (*ortho*-phenylenediamine [OPDA]) was carried out in our labs in a way of synthesizing mordenite zeolite with varying pore geometry and high Si/Al ratio [11].

Titanium atoms incorporated into the zeolite framework serve as catalytic sites thus; the content of framework titanium in a zeolite has presented higher activity in performing some catalytic reactions [12–15].  $\text{TiO}_2$  in mixed phase materials are the best photocatalyst reported so far [16]. However, poor adsorption and low surface area properties lead to great limitations in exploiting the photocatalyst to the best of its photoefficiency. Supported  $\text{TiO}_2$  is commonly reported to be less photoactive due to the interaction of  $\text{TiO}_2$  with the support during the thermal treatments [17]. Several attempts have been made to improve the photoefficiency of titania by adding adsorbents like silica, alumina, zeolites, clays, and active carbon [18–23]. Zeolite based photocatalysis is increasingly attracting many researchers and not yet well explored and the work reported so far involved the

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use of high TiO<sub>2</sub> loadings [24]. On the other hand, the higher dispersion state of Ti oxide species inside zeolite frameworks are prerequisites to effective photocatalytic reactions such as reduction of CO<sub>2</sub> by H<sub>2</sub>O and NO decomposition into N<sub>2</sub> [25,26]. Nanoscale anatase materials are also reported to be highly active probably due to high surface areas rather than intrinsic improvement in reactivity [27,28].

In a way of designing an efficient local structure for hosting TiO<sub>2</sub> photocatalyst to control the activity, mordenite zeolite was synthesized and used as a support for TiO<sub>2</sub>; that can be excited under UV-irradiation forming charge-transfer excited state thus involving an electron transfer from O<sub>2</sub><sup>-</sup> to Ti<sup>4+</sup> [29]. Accordingly, the preparation of Ti-mordenite by three different methods namely in situ incorporation during synthesis, impregnation and solid-solid interaction were carried out. These samples have been characterized by thermal analysis, X-ray diffraction, N<sub>2</sub> adsorption, FT-IR, UV-vis and pyridine-FT-IR techniques. The photocatalytic activity of the samples towards degradation of *meta*-chlorophenol was studied; using HPLC, together with revealing the performance of the catalysts, and the possible reaction mechanism.

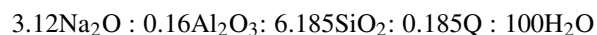
## 2. Experimental

### 2.1. Materials

The materials used were: silicic acid powder, sodium hydroxide pellets (A.R. 98%), aluminum sulfate [Merck, Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>·16H<sub>2</sub>O], *o*-phenylenediamine (Merck), acetyl acetone and titanium tetrachloride (BDH 99% TiCl<sub>4</sub>) and commercial H<sub>2</sub>SO<sub>4</sub>.

#### 2.1.1. Preparation of mordenite

The hydrogels of the following oxide molar compositions were prepared for the synthesis of mordenite zeolite.

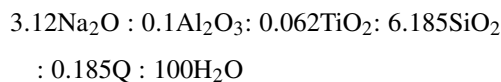


where Q is *o*-phenylenediamine (OPDA) template.

A known amount of NaOH was added to silicic acid/H<sub>2</sub>O system while stirring, followed by heating at 80 °C until clear solution was obtained. The OPDA was dissolved in 10 ml of H<sub>2</sub>O followed by heating at 50 °C for 20 min. Then, the solution of OPDA was added to that of sodium silicate while stirring for 15 min. The aluminum sulfate, on the other hand, was dissolved in calculated amount of H<sub>2</sub>O with adding 0.05-ml concentrated H<sub>2</sub>SO<sub>4</sub>. To the latter solution, the combined solution of sodium silicate and OPDA was added followed by stirring for 30 min. The pH of the mixture was adjusted to 11 by using NaOH (0.1 M) and H<sub>2</sub>SO<sub>4</sub> (0.1 M) solutions. Finally, the mixture was hydrothermally treated at 160 °C in an oil bath, using stainless steel autoclaves for 6 days. The autoclaves were removed at the prespecified time from the oil bath and were quenched immediately with cold water. The solid product was filtered and washed with distilled water until the pH of the filtrate dropped to 8. The products were dried at 110 °C for 10 h then calcined at 550 °C for 6 h in an air oven. This sample was referred to as M.

#### 2.1.2. Preparation of Ti-mordenite (built-in)

The synthesis procedure is described in molar compositions as follows:



The calculated amounts of sodium hydroxide and silica were added to deionized water and stirred with heating at 80 °C until the silica was dissolved. Another clear solution of OPDA was prepared into which a calculated amount of TiCl<sub>4</sub> was added parallel with 2 ml of acetyl acetone followed by vigorous stirring at room temperature for 15 min. The aluminum sulfate, on the other hand, was dissolved in calculated amount of H<sub>2</sub>O with adding 0.05-ml concentrated H<sub>2</sub>SO<sub>4</sub>. A simultaneous addition of the silicate solution was carried out into the complex (composed of OPDA containing TiCl<sub>4</sub> and aluminum sulphate) followed by stirring for 30 min till homogenization of the mixture was achieved. Then, the reaction mixture was transferred to a 300 ml stainless-steel autoclave and maintained in the oil bath at 180 °C, for the crystallization period of 10 days under autogeneous pressure. The Autoclave was removed from the oil bath and quenched under cold water for product identification. The solid product was filtered and washed with distilled water until the pH of the filtrate dropped to 8. The solid product was then dried at 120 °C for 10 h and calcined at 550 °C for 6 h. This sample was referred to as TiM<sub>in</sub>.

#### 2.1.3. Preparation of TiO<sub>2</sub> in mordenite by solid-solid interaction

This sample was prepared by the mechanical mixing of mordenite that synthesized in this work of  $S_{\text{BET}} = 324 \text{ m}^2/\text{g}$  with TiO<sub>2</sub> sample. Mordenite was degassed at 300 °C for 3 h (under a reduced pressure of 10<sup>-5</sup> Torr) prior to mechanically mixing with TiO<sub>2</sub> (prepared by hydrolysis of TiCl<sub>4</sub> using NH<sub>4</sub>OH, washed with distilled water, heated at 110 and finally calcined at 550 °C) then left under stirring in a schlink tube overnight under a flowing nitrogen gas at a rate of 30 cm<sup>3</sup>/min. The sample was evacuated in a conventional high vacuum manifold (~10<sup>-5</sup> Torr) system while the temperature was slowly increased at a ramping rate of 0.33 °C/min till 300 °C at which the temperature was kept constant for 4 h. The prepared sample gave TiO<sub>2</sub> loading 6.5 on the bases of molar ratio. The sample was calcined at 550 °C, in air for 6 h. The obtained sample was referred to as TiM<sub>ss</sub>.

#### 2.1.4. Preparation of TiO<sub>2</sub> in mordenite by impregnation method

This sample was prepared by impregnation of mordenite that synthesized in this work. Na-mordenite was stirred with a solution of TiCl<sub>4</sub> to give TiO<sub>2</sub> loading of 6.5 on the bases of molar ratio. The sample was dehydrated in an oven 110 °C for 6 h and was eventually calcined at 550 °C for 6 h. The obtained sample was referred to as TiM<sub>imp</sub>. TiO<sub>2</sub> purchased from Degussa (P-25 having 85% anatase and 15% rutile of surface area 50 m<sup>2</sup>/g) was used for comparison purposes.

## 2.2. Characterization techniques

The X-ray diffraction patterns of various zeolite samples were measured by using a Philips diffractometer (type PW 3710) equipped with Cu K $\alpha$  radiation. The samples were measured in the  $2\theta$  range of 10–50° at 30 kV and 10 mA with a scanning speed of  $2\theta = 2.5^\circ \text{ min}^{-1}$ . The crystal sizes of the prepared materials were determined using the Scherrer equation. The instrumental line broadening was measured using a LaB6 standard. The crystallinity of the modified samples was calculated using the ratio of the sum of the areas of the most intense peaks for the prepared samples ( $2\theta = 16.8, 19.6, 22.2, 28.2$  and  $32.3 \text{ \AA}$ ) to that of the same peaks of as-synthesized Na-mordenite; of this work, and multiplying by 100. Rietveld method, which permits the reproduction of the whole diffractogram through the optimization of both structural (peak position and intensity) and non-structural (peak shape) data is used to evaluate the determination of cell parameters and volume increase due to insertion of guest ions in the zeolite structure.

In situ FT-IR spectra of the samples were recorded with JASCO single beam FT-IR 5300 spectrometer with 50 co-added scans at  $2 \text{ cm}^{-1}$  resolution. The sample was pressed into a self-supporting wafer and mounted in a quartz infrared cell with CaF<sub>2</sub> windows that connected to vacuum manifold ( $10^{-5}$  Torr) constructed of Pyrex glass with a dead volume of  $301 \text{ cm}^3$ . The infrared cell was equipped with an electric furnace and the sample temperature was adjusted by using the temperature controller connected to a thermocouple made of nickel chrome. The IR sample was prepared by pressing the catalyst powder ground in an agate mortar to a wafer of ca.  $30 \text{ mg cm}^2$  and then outgassing at  $300^\circ \text{C}$  for 3 h before exposure to reactant gases. IR measurements were all carried out at room temperature. All of the spectra presented were obtained by subtraction of the corresponding background reference.

The IR spectral changes of pyridine adsorption were measured in the  $1700\text{--}1400 \text{ cm}^{-1}$  region. Pyridine (10 Torr) was admitted into the cell and equilibrated with the sample for 30 min. Excess pyridine was then pumped out before recording the spectra: pyridine was pumped out at 25 and  $100^\circ \text{C}$  each time.

UV–vis diffuse reflectance spectra of the samples were measured using a JASCO V-570 unit, serial no. (29635), at scanning speed  $4000 \text{ nm/mm}$  and a bandwidth  $2 \text{ nm}$ . The samples were measured in the wavelength range from 200 to 500 nm. The samples were prepared as self-supporting wafers and were recorded at room temperature.

TG and DTA thermograms were performed for uncalcined samples (10 mg) using Model 50 Shimadzu units in the range of  $25\text{--}1000^\circ \text{C}$  with a temperature rise of  $10^\circ \text{C/min}$ . Dry nitrogen was used as a purge gas at a rate of  $30 \text{ ml/min}$ .

The nitrogen adsorption isotherms were measured at  $-196^\circ \text{C}$  using a conventional volumetric apparatus. The specific surface area was obtained using the BET method. The micropore volume and the external surface area were obtained from the  $t$ -plot method.

Experiments in the darkness were performed to determine the amounts of MCP adsorbed on different TiO<sub>2</sub>-mordeites. The

adsorption capacity of *m*-chlorophenol (MCP) was measured with adjusting the pH at 6.5. In 250 ml of MCP solution at initial concentration of  $5 \times 10^{-3} \text{ M}$ , 0.1 g of samples were added and then agitated in the darkness for 30 min. The suspensions were centrifuged to remove TiO<sub>2</sub> particles and the retained concentration of MCP,  $C$  was determined. The amount of MCP adsorbed onto the TiO<sub>2</sub>-zeolites was determined by the difference between the initial and the retained concentration of MCP divided by the initial concentration of MCP:

$$\text{uptake \%} = \frac{C_0 - C}{C_0} \quad (1)$$

The photoreactivity experiments were carried out in a cylindrical Pyrex glass reactor containing 0.10 g of catalyst and 250 ml of aqueous solution of *meta*-chlorophenol (MCP). The concentration of the organic compound was  $5 \times 10^{-3} \text{ M}$ . A 125 W medium pressure Hg lamp (312 nm) immersed within the photoreactor was used and the mean value of the incident radiation power, determined using an UV (Digital UVX 36) was  $13.5 \text{ mW cm}^2$ . The Products were analyzed by HPLC using Dionex p580 pump equipped with a Dionex column DX-120 by means of ion chromatography Dionex-pac attached with AS14 column. Chloride ions were determined through using LiChrospher RP-18 column ( $4 \text{ mm} \times 125 \text{ mm}$ ). Acetonitrile was used as eluent at  $1 \text{ ml/min}$ .

## 3. Results and discussion

### 3.1. Thermal analyses

TG and DTA thermograms of uncalcined M and TiM<sub>in</sub> samples are shown in Fig. 1; in the temperature range from 25 to  $1000^\circ \text{C}$ . TG analyses indicate that the total weight loss of TiM<sub>in</sub> (17.5%) is exceeding that of M (14%). The low temperature weight losses are associated with the removal of water while the high temperature ones are due to the decomposition of the template OPDA. Correspondingly, two endothermic peaks at 128 and  $153^\circ \text{C}$  are observed in the DTA curves for M and TiM<sub>in</sub>, respectively. This indicates that the interaction between water molecules and titano-aluminosilicate is stronger than that with aluminosilicate structure. This could give a hint about the microporosity of the former comparatively. Varying the decomposition temperatures of the template for M ( $322$  and  $537^\circ \text{C}$ ) and TiM<sub>in</sub> ( $402, 496$  and  $552^\circ \text{C}$ ) samples indicate that the interaction of template has different strengths within the two samples. The weight loss expressing OPDA decomposition, on the bases of sharing four molecules per unit cell (4.75%), indicates lower amount in M (3.71%) and higher amount in TiM<sub>in</sub> (7.86%). In addition, exceeding the endothermic maxima appeared at 402 and  $552^\circ \text{C}$  in TiM<sub>in</sub> than those seen in M ( $322$  and  $537^\circ \text{C}$ ) verified the strong association of the template within titanoaluminosilicate structure comparatively. This means that OPDA molecules will find some restrictions during decomposition in the narrow pores present in TiM<sub>in</sub> ( $552^\circ \text{C}$ ) than that in M ( $537^\circ \text{C}$ ). This was in agreement with the observed decrease in weight loss of TiM<sub>in</sub> when compared with that of M in the temperature range  $510\text{--}570^\circ \text{C}$ . However, the observed increase

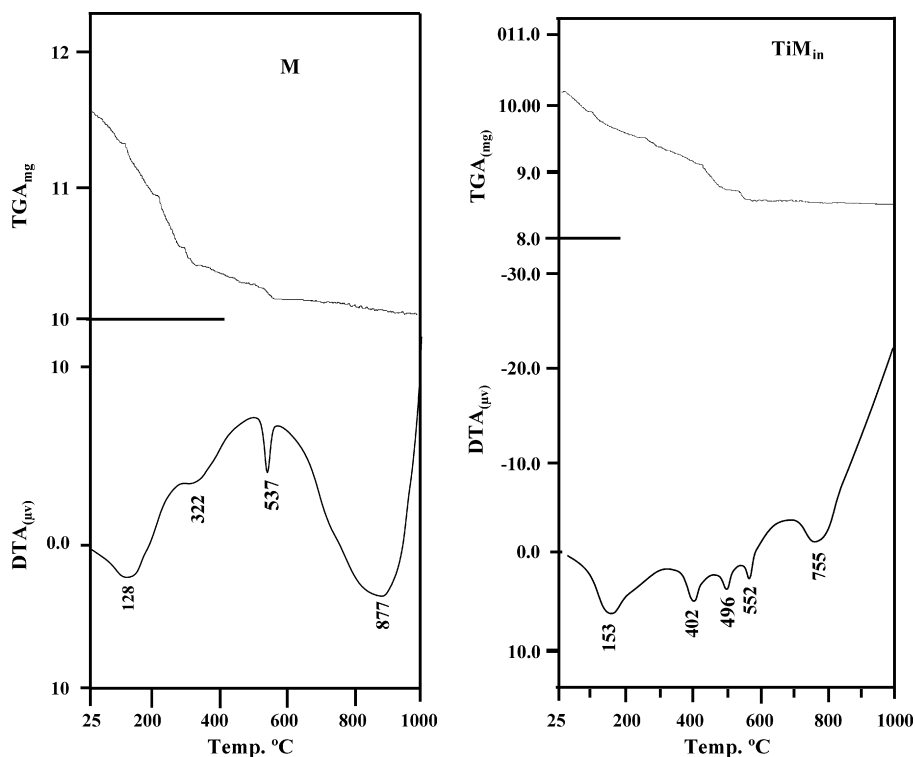


Fig. 1. TGA and DTA curves of uncalcined mordenite and Ti-mordenite samples synthesized by *ortho*-phenylenediamine template.

in weight loss of  $\text{TiM}_{\text{in}}$  in the 300–510 °C range than that of M implies also the presence of wider pores in  $\text{TiM}_{\text{in}}$  indicating that it acquires micro-mesoporous structure. This was in harmony with the total pore volume results that showed an enhancement in  $\text{TiM}_{\text{in}}$  than that in M as a result of Ti incorporation. Ti incorporation in M was responsible for the earlier appearance of the endothermic peak at 755 that originally seen at 877 °C in M, confirming the lower thermal stability of the former comparatively.

### 3.2. X-ray diffractograms

Fig. 2 shows the XRD patterns of various Ti/mordenites synthesized by the three methods, in comparison with that of the parent mordenite synthesized in this work. The former patterns showed quite similar diffraction lines to that of the latter reflecting that they all have mordenite structure. However, minor variations were revealed, for example,  $\text{TiM}_{\text{in}}$  indicates small lines due to titanium-silicate; at  $2\theta = 20.08, 26.31, 28.40$  and  $30.44^\circ$  [30], where  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{ss}}$  samples exhibit  $\text{TiO}_2$

anatase lines at  $2\theta = 37.70, 38.45$  and  $47.87^\circ$  in addition to non-exposed lines of same species in the range of  $2\theta = 25\text{--}28^\circ$  due to exceeding the intensities of mordenite lines in this range rather than a decrease. No peaks for crystalline titania in  $\text{TiM}_{\text{in}}$  can be identified indicating probably the well incorporation of titania particles. Of particular interest, a shift into higher values of  $d$  for M was perceived in  $\text{TiM}_{\text{in}}$  exceeding  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{ss}}$  samples signifying the well incorporation of Ti in the framework of the former. The  $\text{TiM}_{\text{in}}$  pattern shows a partial loss in crystallinity with respect to the M sample where  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{ss}}$  samples show a marked decrease.

The lattice parameters ( $a, b, c$ ) and unit cell volume ( $V$ ) of Ti-mordenites are summarized in Table 1. Upon comparison, the unit cell parameters and volume of the  $\text{TiM}_{\text{in}}$  sample revealed an enhancement when compared with those of  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{ss}}$ , advocating the successful incorporation of Ti into the framework of mordenite in  $\text{TiM}_{\text{in}}$  replacing Si atoms that owned lower bond distances with oxygen (0.160 nm) than those of Ti–O (0.182 nm). Accommodating small titania particles in the micropores caused an expansion in unit cells and volume

Table 1  
Effect of different preparation methods of Ti-mordenites on crystallites size, crystallinity percentages and unit cell parameters

Sample	Crystallite size $D$ (nm)	Cell parameters ( $\text{\AA}$ )			Cell volume $V$ ( $\text{\AA}^3$ )	Crystallinity (%)
		$a$	$b$	$c$		
Mordenite	48.40	18.129	20.413	7.387	2733.69	100
$\text{TiM}_{\text{in}}$	51.63	18.162	20.604	7.530	2817.77	86
$\text{TiM}_{\text{ss}}$	44.02	17.922	20.466	7.528	2761.10	62
$\text{TiM}_{\text{imp}}$	49.48	18.029	20.009	7.761	2799.92	53

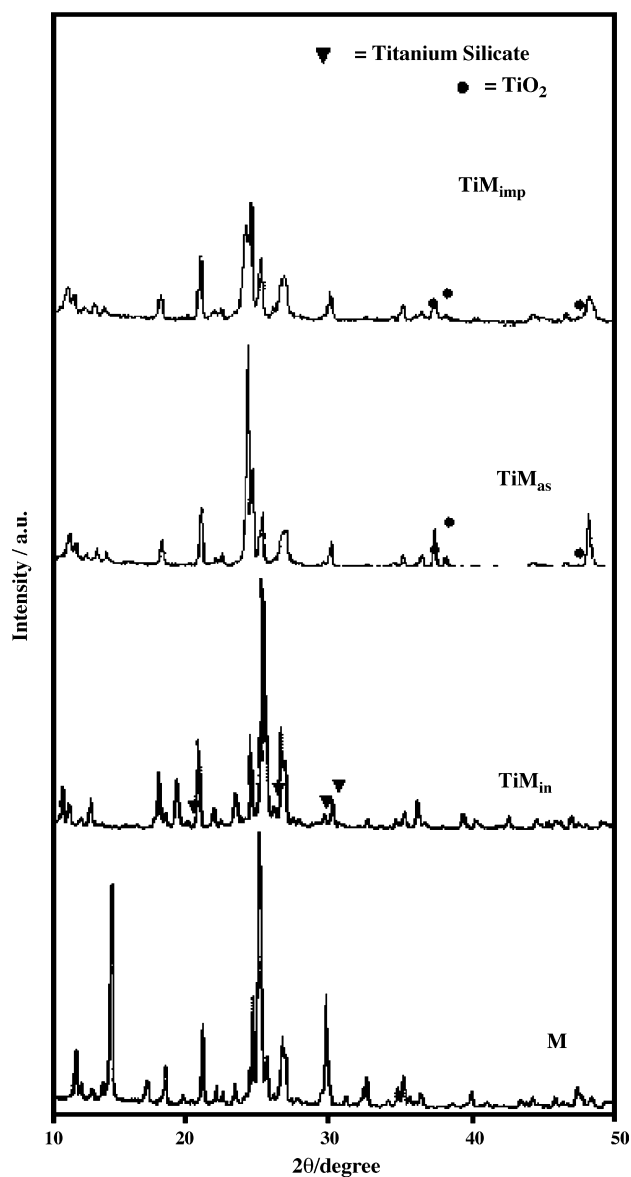


Fig. 2. X-ray diffraction patterns of mordenite and Ti-mordenites prepared by different methods.

for  $\text{TiM}_{\text{in}}$  and exceeded those of post-synthesis ones ( $\text{TiM}_{\text{ss}}$  and  $\text{TiM}_{\text{imp}}$ ). One can notice that the strong interaction of  $\text{TiCl}_4$  with M in  $\text{TiM}_{\text{imp}}$  was higher than that of  $\text{TiO}_2$  with M in  $\text{TiM}_{\text{ss}}$  in the sense of decreasing the crystallinity of  $\text{TiO}_2$  species in the former comparatively.

Table 2

Some Surface characteristics of mordenite and  $\text{TiO}_2$  loading by different preparation methods, heating at  $300^\circ\text{C}$  under a reduced pressure of  $10^{-5}$  Torr

Sample	$S_{\text{BET}}$ ( $\text{m}^2/\text{g}$ )	$S_i$ ( $\text{m}^2/\text{g}$ )	$V_{\text{p}}^{\text{total}}$ ( $\text{cm}^3/\text{g}$ )	$r^-$ ( $\text{\AA}$ )	$S^\mu$ ( $\text{m}^2/\text{g}$ )	$S^{\text{ext}}$ ( $\text{m}^2/\text{g}$ )	$S^{\text{wid}}$ ( $\text{m}^2/\text{g}$ )	$V_{\text{p}}^\mu$ ( $\text{cm}^3/\text{g}$ )	$V_{\text{p}}^{\text{wid}}$ ( $\text{cm}^3/\text{g}$ )	Microporosity (%)
Mor.	351	286	0.316	23	286	66	30	0.257	0.05892	81
$\text{TiM}_{\text{in}}$	384	371	0.347	23	327	58	26	0.295	0.05198	85
$\text{TiM}_{\text{ss}}$	280	262	0.319	29	226	54	31	0.258	0.06123	81
$\text{TiM}_{\text{imp}}$	514	442	0.418	20	440	74	30	0.358	0.05983	86

$S_{\text{BET}}$ : total surface area determined using BET equation,  $S_i$ : specific surface areas determined from  $V_{I-t}$  plots of different adsorbents.  $S^{\text{wid}}$ : wide pore surface area,  $S^\mu$ : micropores surface area,  $V_{\text{p}}^{\text{wid}}$ : the volume adsorbed in wide porous,  $V_{\text{p}}^\mu$ : micropore volume,  $r^-$  ( $\text{\AA}$ ): average pore radius,  $V_{\text{p}}^{\text{total}}$ : the total pore volume at  $p/p^0 = 0.95$ ,  $S_{\text{ext}}^{\text{t}}$ : external surface area of microporous.

### 3.3. Texture assessment

The results of  $\text{N}_2$ -adsorption data obtained for all samples including the specific surface area,  $S_{\text{BET}}$ ; total pore volume,  $V_{\text{p}}$ ; average pore radius,  $r^-$ ; micro and mesopores volumes and surfaces  $V_{\text{p}}^\mu$ ,  $V_{\text{p}}^{\text{wid}}$ ,  $S^\mu$  and  $S^{\text{wid}}$  are collected in Table 2. Inspecting this Table, an increase in  $S_{\text{BET}}$  and  $V_{\text{p}}$  is produced for  $\text{TiM}_{\text{in}}$  and  $\text{TiM}_{\text{imp}}$  samples when compared with that of the parent M. However, the  $\text{TiM}_{\text{ss}}$  sample indicates a decrease in  $S_{\text{BET}}$  (20%),  $V_{\text{p}}$  does not, and an increase in  $r^-$ , that was the largest value among all samples (29  $\text{\AA}$ , Table 2). Accommodation of small titania particles in the micropores of M in  $\text{TiM}_{\text{ss}}$  could be responsible for the devoted expansion of the pore radius. The external surface area  $S^{\text{ext}}$  of  $\text{TiM}_{\text{in}}$  (58  $\text{m}^2/\text{g}$ ), calculated from the slope of the  $V_{I-t}$  curve, is lower than that of M (66  $\text{m}^2/\text{g}$ ), confirming that  $\text{TiM}_{\text{in}}$  preserve higher microporous texturing. This microporosity was also emphasized by the percentage microporosity ( $V_{\text{p}}^\mu/V_{\text{p}} \times 100$ ) that reached 85% (Table 2). Thus, one can attribute the small increase in  $S_{\text{BET}}$  of this sample (9.4%) when compared with M to titaniumsilicate species that has been emphasized by XRD investigations.

The sample  $\text{TiM}_{\text{ss}}$  possessed the lowest  $S_{\text{BET}}$  among all samples. This agrees with the greater  $r^-$  value of the sample. The high value of  $V_{\text{p}}^{\text{wid}}$  for  $\text{TiM}_{\text{ss}}$  also confirms the evolution of mesoporosity. Thus, the texture of sample  $\text{TiM}_{\text{ss}}$  appears more open than rest of the samples (wide), as  $\text{TiM}_{\text{ss}}$  possesses the highest  $r^-$  value, and also deep as that of M (equal  $V_{\text{p}}$  values).

On the other hand, the  $\text{TiM}_{\text{imp}}$  sample presented the highest  $S_{\text{BET}}$  (514  $\text{m}^2/\text{g}$ ) as the micropores are characterized by a high surface area. The microporosity of the  $\text{TiM}_{\text{imp}}$  sample is additionally confirmed from the percentage microporosity and reached 86% ( $V_{\text{p}}^\mu/V_{\text{p}} \times 100$ ) exceeding rest of the samples (Table 2). This sample as is depicted from its  $V_{I-t}$  plot (not shown) indicates pore narrowing character; that was confirmed from the decreased value of  $r^-$ , resulted from blocking of some pores, the implication is that some pore narrowing has taken place in wide pores during calcination process giving that compensation effect. Growth of titania nanophase was clearly revealed on  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{ss}}$  samples when compared with  $\text{TiM}_{\text{in}}$ , as confirmed previously by XRD investigations.

### 3.4. IR spectra of various samples

The FT-IR spectra of Ti-mordenites and M samples are shown in Fig. 3. The IR spectrum of M shows bands at 1220, 1070, 791, 620, 563 and  $455\text{ cm}^{-1}$ , which are assigned to different vibra-

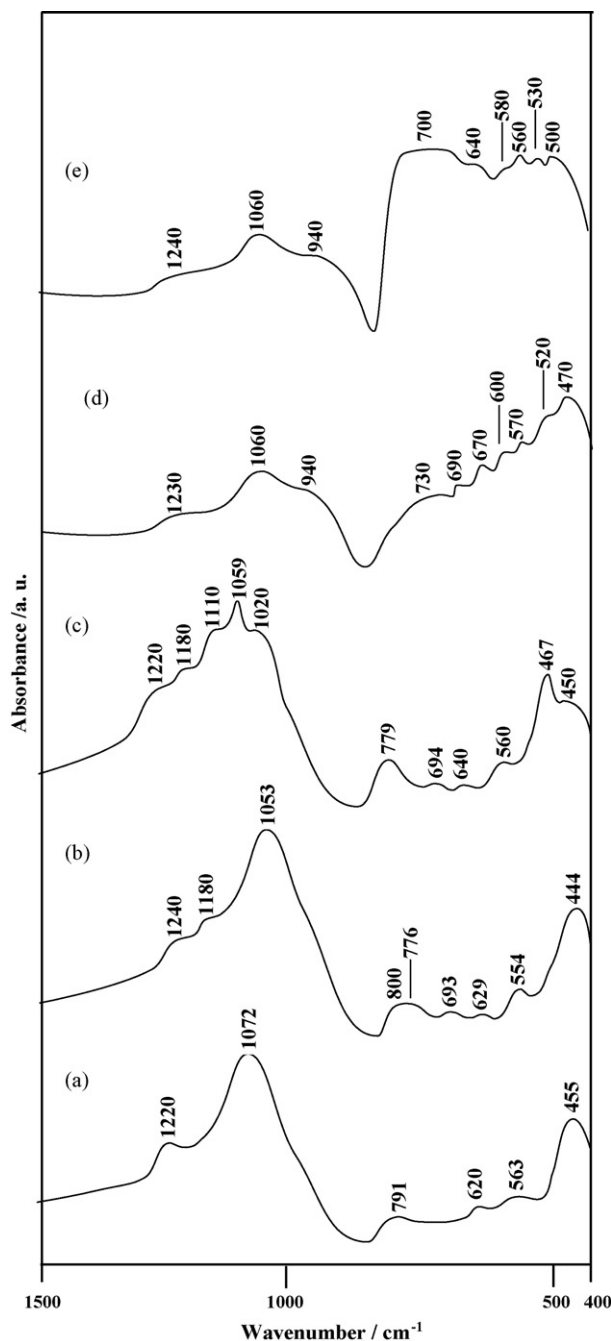


Fig. 3. FT-IR spectra of mordenite and Ti-mordenite samples prepared by different methods: (a) M; (b)  $\text{TiM}_{\text{in}}$ ; (c)  $\text{NH}_4\text{NO}_3$  treated  $\text{TiM}_{\text{in}}$ ; (d)  $\text{TiM}_{\text{imp}}$ ; and (e)  $\text{TiM}_{\text{ss}}$ .

tions of tetrahedral and framework atoms in mordenite zeolite [31]. The IR spectrum of  $\text{TiM}_{\text{in}}$  was similar to that of M except shifting the band at 1072 into 1053  $\text{cm}^{-1}$ ; advocating the well incorporation of titania, as well as appearing a band at 693  $\text{cm}^{-1}$  and a shoulder at 1180  $\text{cm}^{-1}$  those do not show up in the spectrum of M. Ti incorporation in the framework does not alter the wavenumber of the zeolite framework band at 563(554)  $\text{cm}^{-1}$  characteristics of secondary building unit, however, in post-synthesis methods ( $\text{TiM}_{\text{ss}}$  and  $\text{TiM}_{\text{imp}}$ ) some changes in band intensity as well as its shape has been occurred in M structure

presuming marked decrease in crystallinities of these samples, as elaborated previously from XRD investigations. In contrast to  $\text{TiM}_{\text{in}}$ , a band at 950(940)  $\text{cm}^{-1}$  due to vibration of Si–O–Ti linkages is observed in  $\text{TiM}_{\text{ss}}$  and  $\text{TiM}_{\text{imp}}$  [32]. Incorporation of titania within zeolite structure in the  $\text{TiM}_{\text{in}}$  sample is identified through splitting the 791  $\text{cm}^{-1}$  band of  $\nu\text{T-O}$  into doublet at 776 and 800  $\text{cm}^{-1}$ .

As a result of perturbation caused by  $\text{Ti}^{4+}$  in the stretching vibration of Si–O bonds, the  $\text{TiM}_{\text{ss}}$  and  $\text{TiM}_{\text{imp}}$  samples reveal some variations such as splitting of the 563(554)  $\text{cm}^{-1}$  band into 560 and 580  $\text{cm}^{-1}$ , in  $\text{TiM}_{\text{ss}}$ , as well as 570 and 600  $\text{cm}^{-1}$ , in  $\text{TiM}_{\text{imp}}$ , that ascribed to the high concentration of defect sites and crystallite strain along a specific crystallographic direction [33]. In addition, the band of Al–O belongs to alternating  $\text{SiO}_4$  and  $\text{AlO}_4$  tetrahedra at 620  $\text{cm}^{-1}$  [33] is shifted to higher wavenumbers; into 640  $\text{cm}^{-1}$  in  $\text{TiM}_{\text{ss}}$  and to 670  $\text{cm}^{-1}$  in  $\text{TiM}_{\text{imp}}$ , presuming a dealumination process. The broadening of the bands at 640 and 670  $\text{cm}^{-1}$ , which usually provoked from interfering amorphous Al (non-framework) species deposited as a result of dealumination with those of Al in the structure, confirm the latter assignment [34].

Obviously, IR spectroscopy gives direct information on the titanium species in zeolite; highly dispersed framework  $\text{TiO}_x$  stimulates the formation of the 940(950)  $\text{cm}^{-1}$  band in  $\text{TiM}_{\text{ss}}$  and  $\text{TiM}_{\text{imp}}$  samples in addition to extra-framework anatase  $\text{TiO}_2$  that gives rise to bands at 520 and 730; in  $\text{TiM}_{\text{imp}}$ , and 500, 530 and 700; in  $\text{TiM}_{\text{ss}}$ . These bands were indeed shown in the IR result of an anatase  $\text{TiO}_2$  sample (not shown) and; ascribed to Ti–O bonds and rather, emphasized by Gao et al. [33] during their studies on titanium silicalite-1 using FT-Raman spectroscopy. The recognized absence of the fingerprint band of  $\text{Si}\cdots\text{O}^{\delta-}\cdots\text{Ti}^{\delta+}$  at 950(940)  $\text{cm}^{-1}$  in  $\text{TiM}_{\text{in}}$  is probably compensated by the shapely band at 1030  $\text{cm}^{-1}$  that has been assigned by others to the same species [29,30]. For figuring out the presence of this band (1030  $\text{cm}^{-1}$ ), washing with  $\text{NH}_4\text{NO}_3$  solution (1 M) was carried out. Hence bands at 1020 and 1110  $\text{cm}^{-1}$  are exposed, as a result of decreasing sodium ions (Fig. 3c) [34]. This implies that the amount of titanium incorporated into the framework increases as the amount of  $\text{Na}^+$  ions decreases. A band at 1125  $\text{cm}^{-1}$  was identified by Zecchina and co-workers [35] during studying IR spectra of TS-1 and taken as a criterion for Ti–O–Si linkages. However, recognizing two bands in our case could be correlated to the varied environment around Ti atoms in the mordenite framework. The shoulder at 1180  $\text{cm}^{-1}$  is intensified after exchanging with  $\text{NH}_4\text{NO}_3$  and rather forms a distinct band. This band is unambiguously associated with a totally symmetric vibration of the  $\text{TiO}_4$  tetrahedron, achieved through in-phase antisymmetric stretching of the four connected Ti–O–Si bridges [36].

### 3.5. UV–vis diffuse reflectance spectroscopy

Fig. 4 shows diffuse reflectance spectra in the UV–vis region for various Ti-mordenite samples, in comparison with that of M sample. The UV–vis spectrum of the parent sample reveals a band at 240 nm correlated with the zeolite structure [37]. The absorption band at about 200–260 nm are due to electron

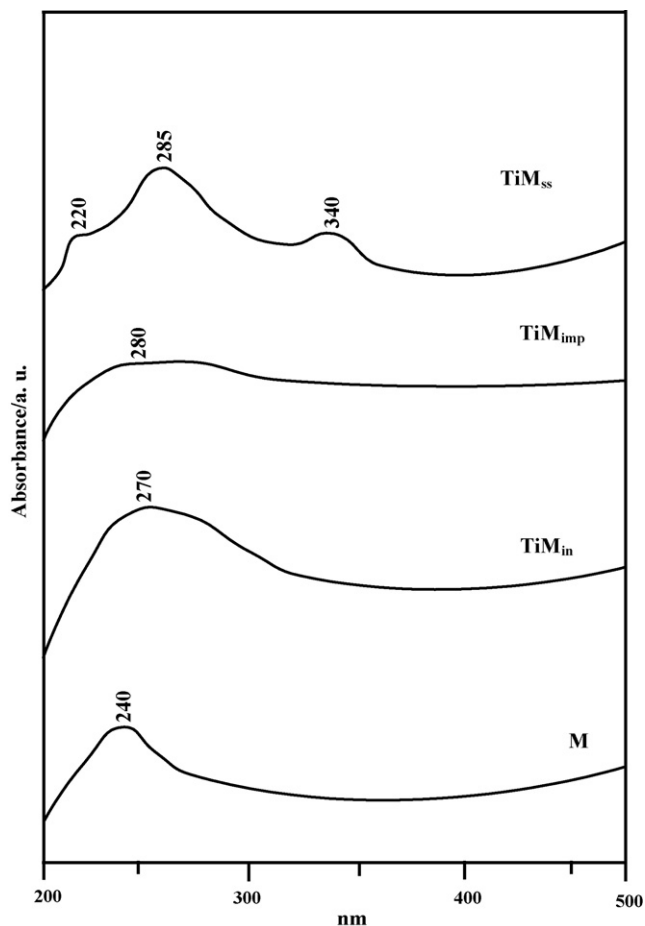


Fig. 4. UV-vis spectra of mordenite and titanium mordenites prepared by different methods.

excitation from ligand oxygen to an unoccupied orbital of the framework  $Ti^{4+}$  [38] in Ti-ligand samples and the maximum at 340 nm, that has been seen only in  $TiM_{ss}$ , are assigned to the absorption band of  $Ti^{4+}$  species in extraframework  $TiO_2$  particles [39]. Concurrently, the  $TiM_{ss}$  sample exposed different Ti species of varying nature and coordination namely tetrahedral  $Ti^{4+}$  ions (220 nm; charge transfer from oxygen to tetrahedral Ti in zeolite framework), highly dispersed Ti species (285 nm) and anatase  $TiO_2$  at 340 nm ( $O^{2-} \rightarrow Ti^{4+}$ ). According to several studies, the broad peak of a maximum at 285 nm; in  $TiM_{ss}$  and at 280; in  $TiM_{imp}$ , can be attributed to the presence of site-isolated Ti atoms in penta- or octahedral coordination [40]. It can also be ascribed to Ti–O–Ti linkages in micro-domain of titania as exhibited previously during synthesis of amorphous  $TiO_2/SiO_2$  that give absorption bands in the range of 260–320 nm [41]. Revealing  $TiO_2$  anatase in XRD results of these samples support the presence of oligomeric  $TiO_2$  species of higher crystallites size in  $TiM_{ss}$  (340 nm) than that in  $TiM_{imp}$ .

The presence of the broad band at 270 nm in  $TiM_{in}$  strongly suggest attributing the major part of Ti into tetrahedral  $Ti^{4+}$  sites, as emphasized from the absence of  $TiO_2$  species in XRD results in addition to developing of some  $Ti^{4+}$  ions in tetrahedral coordination with silica forming Ti–O–Si linkages as deviated from IR outcomes.

### 3.6. Pyridine adsorption

Fig. 5 shows the FT-IR spectra of pyridine (Py) adsorption on various Ti-mordenite samples in comparison with that of M at 25 °C and following evacuation at 100 °C. The spectrum of Py/M at 25 °C exhibits a strong band at  $1444\text{ cm}^{-1}$  due to Lewis pyridine (Lpy) and a weak band at  $1555\text{ cm}^{-1}$  due to Brönsted pyridine (Bpy). The display of the band at  $1645\text{ cm}^{-1}$  is accounted for Bpy species bound probably to strong Brönsted acid sites, besides the ones at  $1612$  and  $1632\text{ cm}^{-1}$ , those no doubt belong to Lewis acid sites. In this essence, one can attribute the small band at  $1650\text{ cm}^{-1}$  to another Bpy species. Evacuation of the sample at 100 °C, indicates a decrease in intensity for the Lpy bands at  $1612$  and  $1632\text{ cm}^{-1}$  in favor of Bpy at  $1650\text{ cm}^{-1}$  with revealing no frequency shifts following the evacuation.

The spectrum exhibited by Py/ $TiM_{in}$  at 25 °C displays the same bands seen by M with an obvious increase in intensity of the Lpy band at  $1442\text{ cm}^{-1}$ . In addition, bands at  $1488$ ,  $1595$ ,  $1660$ ,  $1682$  and  $1694\text{ cm}^{-1}$  were also detected implying an oxidative breakdown of some Lpy into carboxylate species (band at  $1595\text{ cm}^{-1}$ ) [42]. On the other hand, the formation of  $\alpha$ -pyridone species ( $\nu C=O$  band at  $1682\text{ cm}^{-1}$ ) reveals the existence of active nucleophilic (basic) OH groups on this sample [43]. Outgassing at 100 °C retained all the bands back with some variations including a decrease in intensity of Lpy ( $1442\text{ cm}^{-1}$ ), an increase in intensity of Bpy ( $1539\text{ cm}^{-1}$ ) as well as emerging of a new band at  $1502\text{ cm}^{-1}$  characteristics of carbonaceous species [44,45].

For the  $TiM_{imp}$  sample, at 25 °C, a smaller amount of acid sites ( $1440$  and  $1545\text{ cm}^{-1}$ ) than  $TiM_{in}$  was observed together with H-bonding Py bands at  $1590$  and  $1475\text{ cm}^{-1}$  those show instability following outgassing at 100 °C. This sample also shows Lpy bands at  $1620$  and  $1635$ , Bpy at  $1655$  as well as  $\alpha$ -pyridone band characterized by  $C=O$  vibrations at  $1685\text{ cm}^{-1}$ . Outgassing at 100 °C stimulates the formation of two different types of Brönsted acid sites ( $1550$  and  $1530\text{ cm}^{-1}$ ) together with revealing an appreciable intensity for the  $1490\text{ cm}^{-1}$  band implying the presence of appreciable acidic sites after evacuation at 100 °C. In concordance with the prevalence of Brönsted acidity, Lpy bands at  $1620$  and  $1635$ , provoked at 25 °C, were vanished and showed instead a shoulder at  $1625\text{ cm}^{-1}$ . This revealed decrease in Lewis acidity was in favor of that for Brönsted ones ( $1530$ ,  $1550$  and  $1645\text{ cm}^{-1}$ ) following evacuation rather than that due to  $\alpha$ -pyridone (oxidation products,  $1680\text{ cm}^{-1}$ ) that suffered a marked decrease. Indeed, this sample shows a higher appreciable Brönsted acidity as well as lower basic sites when compared with those on  $TiM_{in}$  sample.

For the  $TiM_{ss}$  sample, at 25 °C, no FTIR bands in  $1539$ – $1550$  and  $1442$ – $1445\text{ cm}^{-1}$  regions, characteristic, respectively, to Brönsted and Lewis acid sites were observed. Only a band at  $1490\text{ cm}^{-1}$  was present. This suggests the decrease in acidic sites of this sample. In addition, this sample shows H-bonded Py bands at  $1436$  and  $1582\text{ cm}^{-1}$ . The latter bands were vanished following evacuation at 100 °C emphasizing their assignments. In the  $1700$ – $1600\text{ cm}^{-1}$  region bands due to Lpy; at  $1608$ ,  $1632$

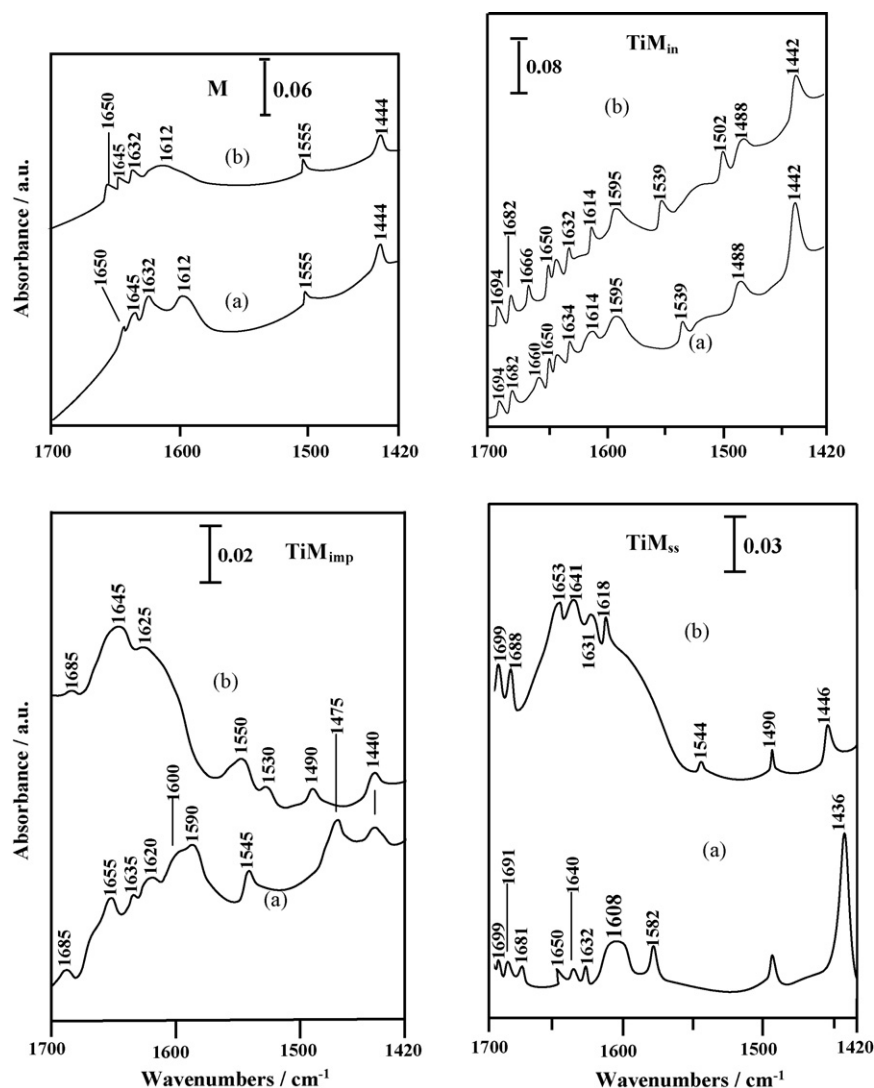


Fig. 5. In situ FT-IR spectra of pyridine adsorption on mordenite and Ti/mordenite prepared by different methods. Pyridine was introduced at room temperature and the spectra were recorded after evacuation for 15 min. (a) Evacuation at 25 °C and (b) evacuation at 100 °C.

and 1640  $\text{cm}^{-1}$ , Bpy; at 1650  $\text{cm}^{-1}$  and oxidative products of pyridine; at 1681, 1691 and 1699  $\text{cm}^{-1}$ , were observed. Evacuation at 100 °C, stimulates the formation of Lpy band at 1446 and Bpy band at 1544  $\text{cm}^{-1}$  as well as enhancing all the bands in the 1700–1600  $\text{cm}^{-1}$  region confined for Lpy, Bpy and oxidative Py species. Following evacuation at 100 °C, a higher amount of Brönsted acid sites is depicted for this sample than  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{in}}$  ones to be in the order:  $\text{TiM}_{\text{ss}} > \text{TiM}_{\text{imp}} > \text{TiM}_{\text{in}} > \text{M}$ . On the other hand, the basic sites on  $\text{TiM}_{\text{ss}}$  were comparable to those on  $\text{TiM}_{\text{in}}$ . The concentration of Lpy in  $\text{TiM}_{\text{ss}}$  (1436  $\text{cm}^{-1}$ ) was found to be higher than those in  $\text{TiM}_{\text{imp}}$  and  $\text{TiM}_{\text{in}}$ . Accordingly, evacuation of Py at 100 °C for  $\text{TiM}_{\text{ss}}$  stimulates the formation of acid–base pair sites.

### 3.7. Activity test

#### 3.7.1. Adsorption of meta-chlorophenol on various samples

Adsorption kinetics of meta-chlorophenol (MCP) on various catalysts were conducted in order to reveal whether or not

the adsorption capacity, beside the other parameters that will be thoroughly investigated, affected the Photocatalytic activity. The adsorption isotherms of MCP on various catalysts, shown in Fig. 6 revealed the followings: (i) the adsorption rate of MCP on mordenite recorded the lower affinity comparatively revolving lower electron accepting capacity, i.e. low Lewis acidity; (ii) among Ti-mordenites,  $\text{TiM}_{\text{in}}$  and  $\text{TiM}_{\text{ss}}$  showed the highest adsorption capacity (25–27%). This is probably caused by complexation between the oxygen lone-pair electrons of phenol and newly developed Lewis acid sites ( $\text{Ti}^{4+}$ ) on mordenite especially  $\text{TiM}_{\text{in}}$  that showed both strong and high density Lewis acid sites, as envisaged using Py adsorption. Indeed, cation exchangeable sites generated when Si is replaced by Ti framework [46] are primarily responsible for the adsorption of MCP; (iii) the order of the adsorption capacity of the samples is  $\text{TiM}_{\text{in}} \approx \text{TiM}_{\text{ss}} > \text{TiM}_{\text{imp}} > \text{M}$ . The adsorption isotherm (not shown) of MCP adsorption amount versus equilibrium concentration of aqueous MCP on all TiM samples were measured at pH 6.5; which is close to the isoelectric point



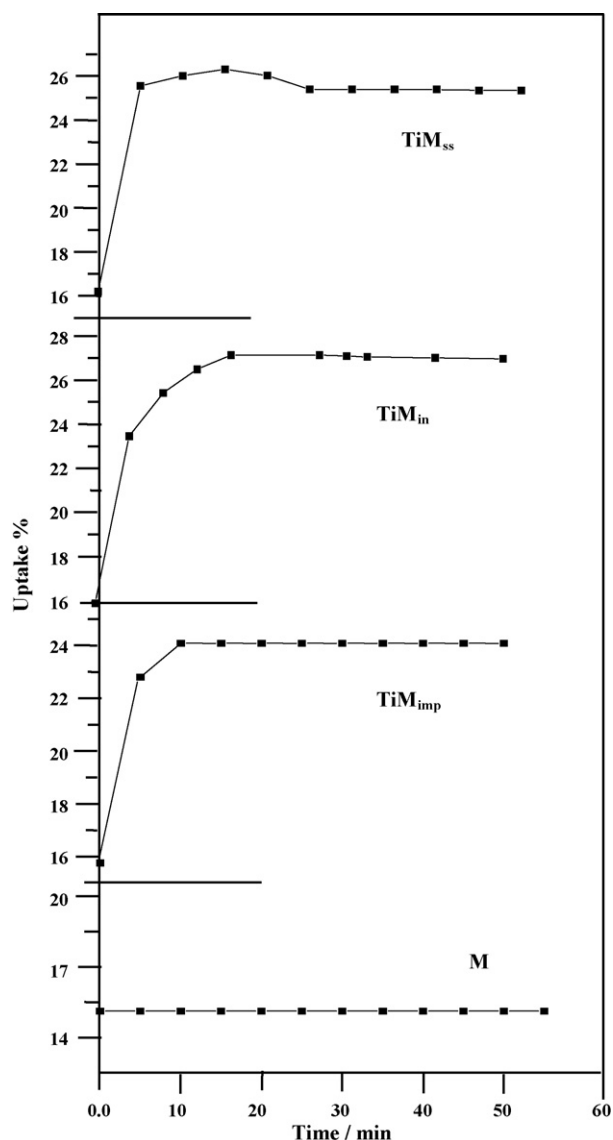


Fig. 6. Sorption kinetics of *m*-chlorophenol on M and TiM samples. Concentration at  $5 \times 10^{-3}$  and temperature  $25^\circ\text{C}$ .

(IEP) of titania ( $\sim 6.1$ ), to nullify the electrostatic interaction between MCP and titania incorporated mordenites. They fit well with Langmuirian type isotherm and the calculated equilibrium constant  $K$  was found to be  $0.15 \times 10^3$ ,  $0.25 \times 10^3$ ,  $0.3 \times 10^3 \text{ m}^3/\text{mol}$  for  $\text{TiM}_{\text{imp}}$ ,  $\text{TiM}_{\text{ss}}$  and  $\text{TiM}_{\text{in}}$ , respectively; (iv) the adsorption saturation was reached in a fewer time on  $\text{TiM}_{\text{imp}}$  than that on  $\text{TiM}_{\text{in}}$ , i.e. forming a plateau, reflecting the facile reaching of MCP into the active sites of the former that showed the highest specific surface area, comparatively. This might indicate that the extent of adsorption is undeniably depend on the surface area of the adsorbent but more importantly depend on the surface polarity of the adsorbent as well as the diffusion of MCP (mass transfer). In this essence, the  $\text{TiM}_{\text{imp}}$  sample presented the lowest pore radius ( $20 \text{ \AA}$ ) between all TiM samples, hindering the transport of the adsorbate molecules when compared with  $\text{TiM}_{\text{ss}}$ , i.e. pore diffusion limitation.

### 3.7.2. Photocatalytic degradation of MCP

Fig. 7 presents the MCP photodegradation kinetics and permits a direct comparison of the performance of the titania modified catalysts. Under the experimental conditions used, the photocatalytic curves follow first-order reaction kinetics. It is well established [47–49] that photocatalysis experiments follow the Langmuir–Hinshelwood model, where the reaction rate  $R$  is proportional to the surface coverage  $\theta$ ,

$$R = -\frac{dC}{dt} = k_r\theta = \frac{k_r KC}{1 + KC}, \quad (2)$$

where  $k_r$  is the reaction rate constant,  $K$  the adsorption coefficient of the reactant, and  $C$  is the reactant concentration. When  $C$  is very small, the  $KC$  product is negligible with respect to unity so that Eq. (2) describes a first order kinetics. The integration of this equation with the limit condition that at the start of irradiation,  $t=0$ , the concentration is the initial one,

$C = C_0$ , gives

$$-\ln\left(\frac{C}{C_0}\right) = K't, \quad (3)$$

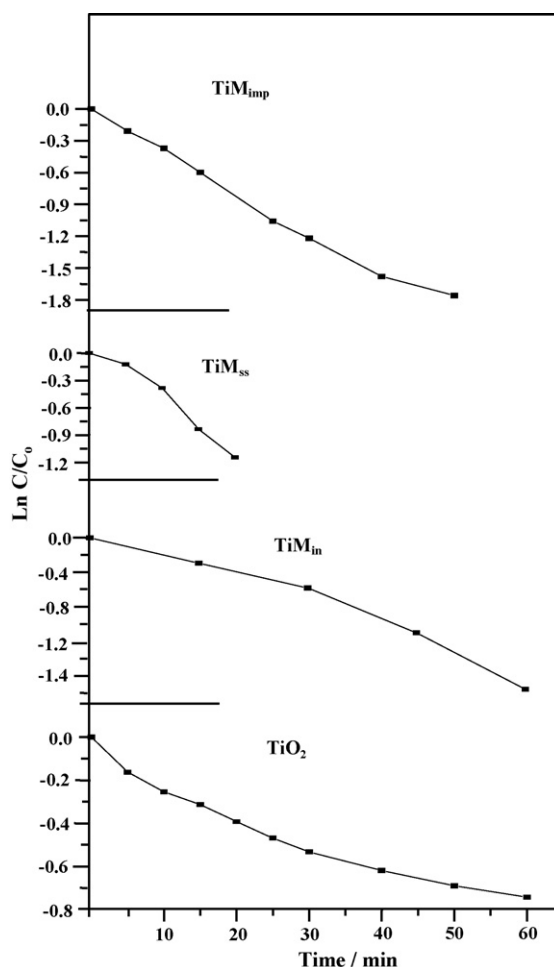


Fig. 7.  $\ln C/C_0$  vs. time of Ti-mordenites catalysts in comparison with  $\text{TiO}_2$ -Degussa.

where  $K'$  is the apparent first-order reaction constant. The kinetic parameter  $K'$  resulting from the application of Eq. (3) indicates the following sequence;  $\text{TiM}_{\text{ss}}$  ( $0.060 \text{ min}^{-1}$ ) >  $\text{TiM}_{\text{imp}}$  ( $0.050 \text{ min}^{-1}$ ) >  $\text{TiM}_{\text{in}}$  ( $0.025 \text{ min}^{-1}$ ) >  $\text{TiO}_2$  (P-25) ( $0.018 \text{ min}^{-1}$ ).

This figure illustrates that mordenites encapsulated  $\text{TiO}_2$  photocatalysts require shorter irradiation time to complete mineralization than bare  $\text{TiO}_2$  due to good adsorption of MCP anions and its intermediates on the zeolite surface where  $\bullet\text{OH}$  radicals are available. The marked higher rate observed for  $\text{TiM}_{\text{ss}}$  (in only 20 min) could be due to the largest pore radius possessed by the catalyst (29 Å) in addition to higher amounts of well-crystallized nanoshaped extraframework  $\text{TiO}_2$  anatase (500, 530, 700  $\text{cm}^{-1}$  as well as 340 nm band) that exceeded ones devoted from  $\text{TiM}_{\text{imp}}$ . The shape of  $\text{TiM}$  curves (except  $\text{TiM}_{\text{ss}}$ ) seems hindered as time elapsed, i.e. the extent of degradation is slowed down, probably due to formation of intermediates and their competitiveness with MCP anions in the photocatalytic degradation process. The difference of the MCP removal between the UV-illuminated and dark (adsorption) conditions ( $\Delta [\text{UV-dark}]$ ) can be used as an apparent measure to evaluate the impact of the UV-illumination on the MCP degradation.  $\text{TiM}_{\text{ss}}$  gives a value equal to 74% ( $[100 - 26]$ ) in a degradation profile extends only to 20 min reflecting an increase in the density of long lived  $\bullet\text{OH}$  radicals comparatively. The possibility of leaching some titanium dioxide from mordenite structure was indeed followed by AAS technique and thus values comprised of 0.05–0.1 ppm were revealed. The loss of titanium into the solution of the mixture could be caused by the formation of hydroperoxy–titanium complex [50]. The amount of leached  $\text{TiO}_2$  from  $\text{TiM}_{\text{ss}}$  was the highest (0.1 ppm) between all catalysts. The  $\text{TiM}_{\text{in}}$  catalyst did not show any  $\text{TiO}_2$  amounts at the end of the reaction suggesting that most of titanium is associated within the zeolite framework. Heterogeneous reactions rather than homogeneous ones are devoted principally on these photocatalysts.

From these results, a highly efficient photocatalytic oxidation of MCP can be achieved using the  $\text{TiM}_{\text{ss}}$  catalyst, which includes highly dispersed anatase  $\text{TiO}_2$  as active species, while dispersed Ti–O where the  $\text{Ti}^{4+}$  was tetrahedrally coordinated are less candidate for MCP oxidations. These results indicate the ability of mordenites in favoring photo-induced electron-transfer reactions and retarding back electron transfer ones [51], i.e. zeolite support delocalizes photo-generated electrons thus prohibiting and/or delaying its recombination with holes.

$\text{Cl}^-$  was released upon degradation of MCP in all tested catalysts and this process was shown to be very fast while using the  $\text{TiM}_{\text{ss}}$  catalyst that showed the highest degradation rate although of the fear of re-adsorbing  $\text{Cl}^-$  on the catalyst surface and thus slowing the rate of formation of reactive  $\bullet\text{OH}$ . At the time of complete disappearance of MCP (within 60 min in the  $\text{TiM}_{\text{in}}$  catalyst (Fig. 8) about 90% of stoichiometric amount of  $\text{Cl}^-$  was recovered suggesting that  $\text{Cl}^-$  containing intermediate compounds are produced in the mineralization process. However, the pH of all mixtures changes from 6.27 to 2.5–2.0 at the end of the reaction. Quite amounts of  $\text{Cl}^-$  react on the other hand, to form HCl, which influences the surface state of titanium containing mordenites and the ionization state of the ionizable MCP molecules. For pH's lower than PZC of titania (6.1), [52] the sur-

face become positively charged and the following equilibrium can be expected:



Since MCP has a negatively charged phenolate function ( $\text{O}^-$ ), its adsorption is favored on a positively charged surface. This result indicates that the acidic solution favors the adsorption of MCP onto the photocatalyst surface. However, the devoted variation in the kinetics of MCP disappearance on various catalysts could be correlated with the extent of  $\text{Cl}^-$  released to intervene with protons to form HCl in short time thus, making the mixture more acidic to further share in enhancing the rate of reaction. In addition, zeolites are known to stabilize reactive intermediate in several photochemical reactions [53] and the reactivity of short chain aliphatic compounds with  $\bullet\text{OH}$  radicals is usually slow. These difficulties have been overcome in  $\text{TiM}_{\text{ss}}$  due to the facile mobility of MCP molecules onto the photoactive  $\text{TiO}_2$  sites present on mordenites surfaces capable of forming reactive  $\bullet\text{OH}$  radicals. On the other hand, crystallites size of  $\text{TiO}_2$  on M in  $\text{TiM}_{\text{ss}}$  is small enough to prohibit scattering of UV light thereby, an electron transfer from excited state of MCP to  $\text{TiO}_2$  on M (in  $\text{TiM}_{\text{ss}}$ ) is generated [54].

Following up the decay of MCP by HPLC for all  $\text{TiM}$  catalysts indicated the same products assuming a similar mechanism. Thus, as an example, the product distribution of the  $\text{TiM}_{\text{ss}}$  catalyst is examined in details (Fig. 8). The primary

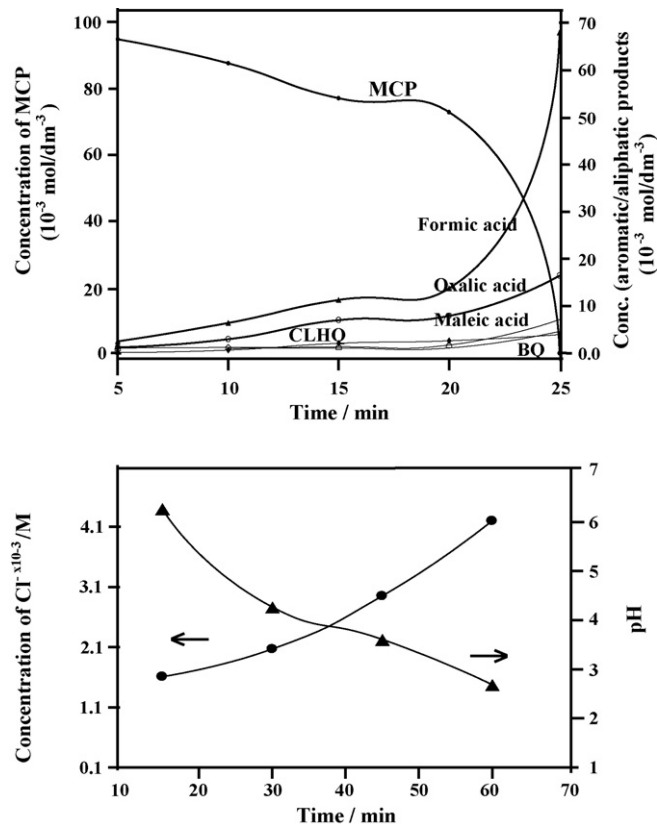


Fig. 8. Photocatalytic degradation of *m*-chlorophenol and formation of  $\text{Cl}^-$  and intermediates on the  $\text{TiM}_{\text{ss}}$  catalyst (0.10 g) as deduced by HPLC and ion chromatography Techniques.

aromatic intermediates identified by HPLC were chlorohydroquinone (CLHQ) and benzoquinone (BQ). Besides, some aliphatic compounds (carboxylic acid and specifically maleic, oxalic and formic acids) were also identified. Since the former products exhibit high UV absorption, their low peak area does not allow their unambiguous identification, however, the latter products, specifically formic acid, appears appreciably at the end of the irradiation time. Thus, the lack of aromatic compounds to be present in significant amount suggests a reasonable efficient ring-opening reaction under the conditions applied. On the basis of our results and literatures devoted for photocatalytic decompositions of *ortho* and *para* chlorophenols [55,56] the following (simplified) mechanism is proposed for the photocatalytic degradation of MCP (Fig. 9). The initial degradation step; following the photochemical generation of  $\bullet\text{OH}$  radicals, is the attack on aromatic ring to yield substituted chlorodihydroxycyclo-hexadienyl radical (1) [57]. Some authors consider that H abstraction may also play an important role (2), and direct electron abstraction from the aromatic compound by  $\bullet\text{OH}$  to give resonance-stabilized cations radical is also

presumed to be of importance [58].  $\bullet\text{OH}$  addition in the *para* position to 3-chloro-phenol, followed by H-abstraction leads to chlorohydroquinone (3). This suggests that a highly favored addition of  $\bullet\text{OH}$  in the *para*-position is formed nullifying those might take place in *ortho* positions due to steric factors. The process is then initiated to form *p*-benzoquinone (4) by realising either  $\text{Cl}^-$  (dechlorinations) or  $\text{HCl}$  formation, as has been confirmed previously. One cannot deny that ring cleavage could be responsible for chlorohydroquinone-benzoquinone transformation. Further, degradation of *p*-benzoquinone with  $\bullet\text{OH}$  and/or  $\text{O}_2$  leads to maleic acid (5) that can be mineralized to  $\text{CO}_2$  via either oxalic (6) or formic acids (7) those indeed strongly accelerated by  $\bullet\text{OH}$  radicals.

To have a clue about the adsorbed species onto the catalyst surface after MCP degradation on  $\text{TiM}_{\text{ss}}$ , an FTIR spectrum was undertaken for the sample following the reaction as illustrated in Fig. 10, in comparison with that of  $\text{TiM}_{\text{ss}}$ , not exposed to the reaction. As it can be seen, oxidized fragments of  $\text{COO}$  groups can easily be identified by the characteristic absorption bands in  $\nu_{\text{s}}(1330\text{--}1480\text{ cm}^{-1})$  and  $\nu_{\text{as}}(1500\text{--}1620\text{ cm}^{-1})$

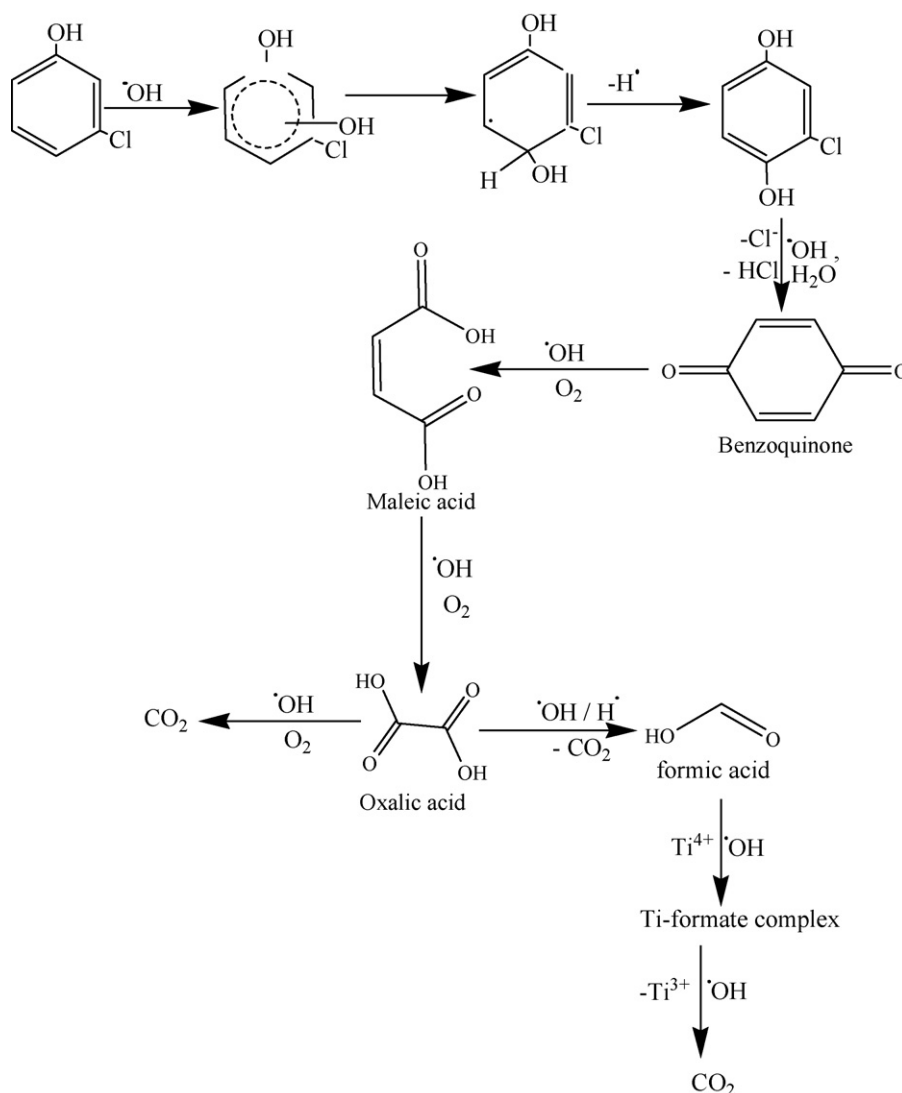


Fig. 9. General reaction pathway proposed for *meta*-chlorophenol mineralization as catalyzed by  $\text{TiM}_{\text{ss}}$  under UV irradiation.

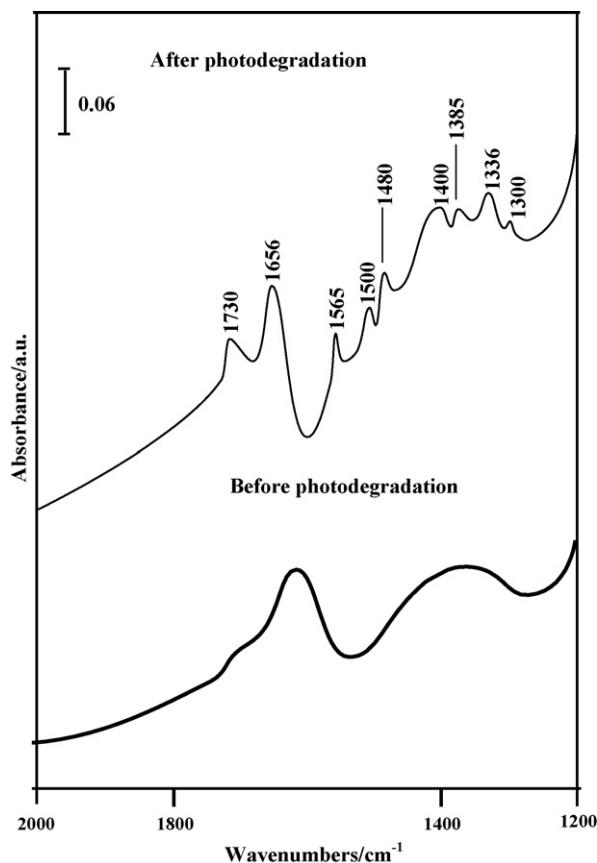


Fig. 10. FT-IR spectra of  $\text{TiM}_{\text{ss}}$  before and after photodegradation of *meta*-chlorophenole.

regions [59]. This indicates the presence of both formate ( $\nu_{\text{as}}\text{HCOO}^-$ :  $1565\text{ cm}^{-1}$ ;  $\nu_{\text{s}}\text{HCOO}^-$ :  $1336\text{ cm}^{-1}$ ) and carboxylate ( $1400\text{--}1480\text{ cm}^{-1}$ ) groups. Furthermore, formic acid vibrational band at  $1730\text{ cm}^{-1}$  characteristics of  $\nu\text{C}=\text{O}$  was also seen together with a small band at  $1385\text{ cm}^{-1}$  due to the  $\delta(\text{C-H})$  vibration. Similar bands at  $\nu_{\text{as}}(\text{HCOO}^-)$ :  $1600\text{--}1560\text{ cm}^{-1}$ ,  $\nu_{\text{a}}(\text{HCOO}^-)$ :  $1370\text{--}1306\text{ cm}^{-1}$  and  $1730(\nu\text{C}=\text{O})$  were depicted when formic acid adsorbed on  $\text{TiO}_2$  and they are assigned to formate and formic acid species, respectively. The presence of C–H stretching vibration bands at  $2840$  and  $2930\text{ cm}^{-1}$  (not shown) emphasize the formation of formate species and in the same time nullifies the possibility of founding any aromatic intermediates due to the absence of C–H stretching band of aromatic moieties over  $3000\text{ cm}^{-1}$ . It can be noted that formic acid and/or formate molecules were the last degraded organic molecules detected before mineralization and found in major amounts than those of oxalic acids. Thus, one might assign the bands occurred in the  $1480\text{--}1400\text{ cm}^{-1}$  range to other oxalate or to adsorbed formate on Ti atoms and/or Na-mordenite.

No significant changes concerning mordenite framework ( $560\text{ cm}^{-1}$ ) were revealed, following the photocatalytic decomposition together with the persistence of the band at  $968\text{ cm}^{-1}$ , ascribed to Ti–O–Si linkages evidencing the stability and integrity of the structure and probably to the non-significant role of latter moieties.

#### 4. Conclusions

The Photocatalytic degradation of *meta*-chlorophenol (MCP) that has been performed by Ti-mordenites in conjunction with UV irradiation was examined. The investigated performance of the catalysts for MCP degradation was directly correlated with their structural and acidic characterizations, which have been thoroughly investigated using XRD, thermal analysis,  $\text{N}_2$  adsorption, FT-IR, UV–vis and pyridine-FTIR techniques. The results were as follows:

1. The  $\text{TiM}_{\text{ss}}$  catalyst exhibited the highest rate of degradation exceeding rest of the samples. This was due to the offered large pore radius ( $29\text{ \AA}$ ) and anatase  $\text{TiO}_2$  species ( $500, 530$  and  $700\text{ cm}^{-1}$  and  $340\text{ nm}$ ). Si–O–Ti linkages were observed in  $\text{TiM}_{\text{ss}}$  [ $950(940\text{ cm}^{-1})$ ] and showed no significant role in the reaction besides, high concentrations of defect sites and crystalline strain as a result of  $\text{TiO}_2$  insertion. The dechlorinations of  $\text{Cl}^-$  as  $\text{HCl}$  increased up the reaction rate tremendously due to decreasing the pH to be lower than the ZPC of  $\text{TiM}_{\text{ss}}$  (from 6.27 to 2.0) thus favoring the adsorption of MCP, which acquired negatively charged phenolate groups. The acidity of  $\text{TiO}_2$  encapsulated inside mordenites helps in accelerating the activity compared with bare  $\text{TiO}_2$  due to improved adsorption of MCP (increasing number of Lewis sites) and delocalization of photogenerated  $e^-/h^+$  by zeolite substrate.
2. Only aliphatic compounds were revealed on final stages of degradation most importantly formates (formic acid) that represented major part of these compounds.  $\text{TiM}_{\text{ss}}$  never showed any changes in structure after performing the reaction reflecting the stability of the catalyst, that indeed needs further investigation to clarify the extent of its stability for industrial application.
3. Incorporating titania inside the framework of zeolite minimizes the photoefficiency of dispersed Ti–O (where  $\text{Ti}^{4+}$  is in tetrahedral structure) as seen in  $\text{TiM}_{\text{in}}$ . Similarly, the strong interaction between  $\text{TiO}_2$  and zeolite also affects the photoefficiency of  $\text{TiO}_2$  as seen in  $\text{TiM}_{\text{imp}}$ . The decreased pore radius, in this sample, hinders the mobility of MCP to the photocatalytic  $\text{TiO}_2$  sites.

#### References

- [1] J.E. Gilbert, A. Mosset, Mater. Res. Bull. 33 (1998) 997.
- [2] P.K. Bajpai, Zeolites 6 (1986) 2.
- [3] G.A. Ozin, A. Kuperman, A. Stein, Angew. Chem. Int. Ed. Eng. 28 (1989) 259.
- [4] D. Cox, T.E. Gier, G.D. Stucky, J. Birlein, J. Am. Chem. Soc. 110 (1988) 2986.
- [5] Y. San, T. Song, S. Qiu, W. Pang, J. Shen, D. Jiang, Y. Yue, Zeolites 15 (1995) 547.
- [6] J. Warzwoda, A.G. Dixon, R.W. Thompson, A. Sacco, J.S.L. Suib, Zeolites 16 (1996) 125.
- [7] S. Qiu, J. Yu, G. Zhu, O. Terasaki, Y. Nozue, W. pang, R. Xu, Microporous Mesoporous Mater. 21 (1998) 245.
- [8] C. Shao, X. Li, S. Qiu, F. Xiao, O. Terasaki, Microporous Mesoporous Mater. 39 (2000) 117.
- [9] M.M. Mohamed, J. Colloid Inter. Sci. 265 (2003) 106.

- [10] T.M. Salama, M.M. Mohamed, I. Othman, G.A. El-shobaky, *Appl. Catal. A* 286 (2005) 85.
- [11] M.M. Mohamed, T.M. Salama, I. Othman, I. Abdallah, *Microporous Mesoporous Mater.* 84 (2005) 84.
- [12] R. Millini, E.P. Massara, G. Perego, G. Bellussi, *J. Catal.* 137 (1992) 497.
- [13] A. Thangaraj, R. Kumar, S.P. Mirajkar, P. Ratnasamy, *J. Catal.* 130 (1991) 1.
- [14] G. Deo, A.M. Turek, I.E. Wachs, D.R.C. Huybrechts, P.A. Jacobs, *Zeolites* 13 (1993) 365.
- [15] E. Astorino, J.B. Peri, R.J. Willey, G. Busca, *J. Catal.* 157 (1995) 482.
- [16] S.O. YaO-Matsuo, M. Ueda, *J. Photochem. Photobiol. A* 168 (2004) 1.
- [17] O. Legrini, E. Oliveros, A. Braun, *Chem. Rev.* 93 (1993) 671.
- [18] J. Matos, J. Laine, J.M. Herrmann, *Appl. Catal. B* 18 (1998) 291.
- [19] N. Takeda, T. Torimoto, S. Sampath, S. Kuwabata, H. Yoneyama, *J. Phys. Chem.* 99 (1995) 986.
- [20] H. Uchida, S. Itoh, H. Yoneyama, *Chem. Lett.* (1993) 1995.
- [21] N. Takeda, N. Iwata, T. Torimoto, H. Yoneyama, *J. Catal.* 177 (1998) 240.
- [22] T. Ibusuki, K. Takeuchi, *J. Mol. Catal.* 88 (1994) 93.
- [23] M.M. Mohamed, T.M. Salama, T. Yanaguchi, *Colloids Surf. A* 207 (2002) 25.
- [24] P.G. Smirniotis, L. Davydov, *Catal. Rev. Sci. Eng.* 41 (1999) 43.
- [25] W.S. Ju, M. Matsuoka, M. Anpo, *Catal. Lett.* 71 (2001) 91.
- [26] M. Anpo, H. Yamashita, Y. Ichihashi, Y. Fujii, M. Honda, *J. Phys. Chem. B* 101 (1997) 2632; M.M. Mohamed, M.M. Al-Esaimi, *J. Mol. Catal. A* 255 (2006) 53; M. Matsuoka, M. Anpo, *J. Photochem. Photobiol. C* 3 (2003) 225.
- [27] R. Bolton, S.R. Cater, in: R.G. Zepp, G.R. Helz, D.G. Crosby (Eds.), *Aquatic and Surface Photochemistry*, Lewis Publishers, Boca Raton, FL, 1994, p. 467.
- [28] T. Sreethawong, Y. Suzuki, S. Yoshikawa, *Int. J. Hydrogen Energy* 30 (10) (2005) 1053.
- [29] T. Ohno, Z. Miyamoto, K. Nishijima, H. Kanemitsu, F. Xueyuan, *Appl. Catal. A* 302 (2006) 62.
- [30] A.M. Velarde, P. Bartl, T.E.W. Nieben, W.F. Holederich, *J. Mol. Catal. A* 147 (2000) 225.
- [31] A. Carati, C. Flego, E. Previde Massara, R. Millini, L. Carluccio, W.O. Parker Jr., G. Bellussi, *Microporous Mesoporous Mater.* 30 (1999) 137.
- [32] A. Trovarelli, G. Dolcetti, C. de Leitenburg, J. Kaspar, P. Finetti, A. Santoni, *J. Chem. Soc. Faraday Trans.* 88 (1992) 1311.
- [33] H. Gao, W. Lu, Q. Chen, *Microporous Mesoporous Mater.* 34 (2000) 307.
- [34] H. Okaniwa, M. Kasagen, *Japan Patent* 2, 149 (1990) 416.
- [35] G. Ricchiardi, A. Damin, S. Bordiga, C. Lamberti, G. Spano, F. Rivetti, A. Zecchina, *J. Am. Chem. Soc.* 121 (2001) 11409.
- [36] S. Bordiga, A. Damin, G. Berlier, F. Bonino, G. Ricchiardi, A. Zecchina, C. Lamberti, *Int. J. Mol. Sci.* 2 (2001) 167.
- [37] M.M. Mohamed, I. Othman, N. Eissa, *Appl. Catal. A* 87 (2005) 93.
- [38] R.J. DAVIS, Z. Liu, T.E. Tabora, W.S. Wielband, *Catal. Lett.* 101 (1995) 34.
- [39] D. Ton, S.V. Nguyen, V. Hulea, E. Dunitriu, S. Kaliaguine, *Microporous Mesoporous Mater.* 57 (2003) 169.
- [40] B. Notari, *Stud. Surf. Sci. Catal.* 37 (1988) 413.
- [41] J. Klass, K. Kulawik, G. Schulz-Ekloff, *Stud. Surf. Sci. Catal.* 84 (1994) 2261.
- [42] M.I. Zaki, G.A.M. Hussein, S.A.A. Mansour, H.A. El Ammawy, *J. Mol. Catal.* 51 (1989) 209.
- [43] M.M. Mohamed, *J. Mol. Catal.* 200 (2003) 301.
- [44] S. Huber, H. Knözinger, *Appl. Catal. A* 181 (1999) 239.
- [45] M.M. Mohamed, *Appl. Catal. A* 367 (2004) 135.
- [46] I. Moriguchi, M. Honda, T. Ohkubo, Y. Mawatari, Y. Teraoka, *Catal. Today* 90 (2004) 297.
- [47] J.M. Hermann, *Catal. Today* 24 (1995) 157–164.
- [48] N.T. Durg, N.V. Khoa, J.-M. Herrmann, *Inter. J. Photoenergy* 7 (2005) 11.
- [49] P. Pichat, in: G. Ert, H. Knozinger, J. Weit Kamp (Eds.), *Hand Book of Heterogenous Photocatalysis*, VCH-Wiley, Weinheim, 1997, p. P-2111.
- [50] J. Grzechulska, A.W. Morawski, *Appl. Catal. B* 36 (2002) 45.
- [51] M.V. Shanker, S. Anandan, N. Venkatachalam, B. Arabindoo, V. Murugesan, *Chemosphere* 63 (2006) 1014.
- [52] I. Ilisz, A. Donbi, K. Mogyorosi, A. Forkas, I. Dekany, *Appl. Catal. B* 39 (2002) 247.
- [53] R. Chatti, S.S. Rayalu, N. Dubey, N. Labhsetwar, S. Devotta, *Solar Eng. Mater. Solar Cell* 91 (2007) 180.
- [54] J.J. Lee, Y. Kim, M. Yoon, *J. Photosci.* 8 (2001) 257.
- [55] R. Saulea, E. Brillas, *Appl. Catal. B* 29 (2001) 135.
- [56] L. Jakob, T.L. Hashem, S. Burki, N.M. Guindz, A.M. Braun, *J. Photochem. Photobiol. Chem. A* 75 (1993) 97.
- [57] M.E. Snook, G.A. Hamilton, *J. Am. Chem. Soc.* 96 (1974) 860.
- [58] A.A. Davydov, *Infrared Spectroscopy of Adsorbed Species on the Surface of Transition Metal Oxides*, John Wiley & Sons, 1990, p. 148.
- [59] J. Arana, O.G. Diaz, M.M. Saracho, J.M. Dona Rodriguez, J.A. Herrera Melian, J.P. Pena, *Appl. Catal. B* 32 (2001) 49.